# Giant shear mode electromechanical coupling in c-axis tilted ScAlN films

c 軸傾斜配向 ScAIN 膜の擬似すべりモードおける巨大圧電性

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#### 1. Introduction

Significant increase of piezoelectricity was recently found experimentally in the Sc heavily doped AlN film [1]. Theoretical prediction based on density-functional theory was also consistent with the experimental results [2]. Fifth fold increase of piezoelectric constant  $d_{33}$  value, extensional compared to the AlN film, was observed in the Sc<sub>0.43</sub>Al<sub>0.57</sub>N alloy films [1]. However, piezoelectric characterization in these studies were based on direct current measurement using a piezo-meter. To use these films in BAW and SAW applications, it is important to know the electromechanical coupling, k values in the microwave frequencies. In addition, it is interesting to investigate the shear mode properties in the ScAlN films.

In this study, we present the quasi-thickness extensional mode and quasi-thickness shear mode electromechanical coupling coefficient  $k_{33}$ ' and  $k_{15}$ ' values in the c-axis tilted ScAlN films.

On the other hand, piezoelectricity in most of the ferroelectric material, for example, PZT and BaTiO<sub>3</sub> deteriorates above 500 °C due to phase transition. Piezoelectricity starts to deteriorate at 600 °C even in LiNbO<sub>3</sub> which possesses high  $T_c$  [3]. ScAlN film have a potential to sustain large piezoelectricity at high temperature. Here, electromechanical properties of the films in 600 °C are also presented and compared with c-axis tilted ZnO films which possess highest k values in the wurtzite.

# 2. Method

# 2.1 c-axis tilted ScAlN films

c-axis tilted ScAlN films  $(1.5-3 \mu m)$  were deposited on Al/silica glass substrate using a planar RF magnetron sputtering system. ScAl alloy metal with 37% Sc concentration was used as a target. Two samples A and B with different c-axis tilt angle and degree of orientation were prepared. An electron probe micro analysis (EPMA) showed that

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Sc/Al composition ratio in the alloy films were in accordance with that in the target alloy.

# 2.2 Crystalline orientation and piezoelectric properties in the film

First, crystalline orientation of the films was investigated by using a 0002 plane pole figure analysis. c-axis tilt angle and the degree of the crystalline orientation were determined from the peak value and FWHM value of the a  $\psi$ -scan profile curve of the 0002 pole.

Next, copper top electrode films were deposited onto film samples, and composite resonator structure were fabricated.  $k_{33}$  and  $k_{15}$  values of the film layer were determined by comparing experimental and theoretical conversion losses of the resonators [4, 5]. Theoretical conversion loss characteristics were calculated using modified Mason's equivalent circuit model including effect of c-axis tilt and electrode layers [6].

In addition, k values were measured at 25-600 °C in air by using heating/cooling stage (Linkam, LK-600PH). Top copper electrode film was removed and high temperature tolerant Pt or Au top electrode film was Dc-sputter-deposited.

# 3. Results

#### 3.1 Crystalline orientation

Figure 1 shows the typical 0002 pole figure of the ScAlN film (sample A). c-axis tilt angle is found to be 19°.  $\psi$ -scan FWHM of pole is measured to be 8.5°, indicating relatively high crystalline orientation. In sample B c-axis tilt angle and  $\psi$ -scan FWHM are measured to be 14° and 6.7°, respectively.

# 3.2 Piezoelectric properties

Figure 2 shows the calculated  $k_{33}$ ' and  $k_{15}$ ' values as a function of c-axis tilt angle  $\gamma$  in pure single crystalline AlN. Physical constant of AlN reported by Ohasi [7] was used in the calculation. We have plotted determined  $k_{33}$ ' and  $k_{15}$ ' values in the Fig. 2. Both values in polycrystalline ScAlN films far exceeded that in single crystalline AlN. Large  $k_{15}$ ' value of 0.35  $k_{15}^2=12$  %) was obtained in sample A.  $k_{15}$ ' values of the AlN increased with increasing c-axis tilt angle. This tendency implies that further high  $k_{15}$  value is expected if 30° tilted film is obtained.



Fig. 1 (0002) pole figure in the ScAlN film (sample A)

#### 3.3 *k* values in high temperature

Figures 3 shows the temperature characteristics of  $k_{33}$ ' and  $k_{15}$ ' values in the c-axis tilted ScAlN samples. Characteristics for c-axis non-doped AlN films [8] and c-axis tilted ZnO film [6] are also displayed in the figures. In the ZnO film, both  $k_{33}$ and  $k_{15}$ ' values started to decrease at 400 °C, probably due to the increase of n-type conductivity [9] caused by increase of oxygen vacancy. k values recovered to the value at room temperature when sample was cooled. In contrast, significant deterioration of k value was not observed in the c-axis tilted AlN and ScAlN films. Large hysteresis characteristic did not appear in all samples. We could not perform the measurement above 600 °C because of the specification limit of the heating stage. Perhaps, ScAlN films can be used in higher temperature.

#### 4. Conclusions

Giant electromechanical coupling  $k_{33}$ ' and  $k_{15}$ ' of 0.35 ( $k_{15}^2=12$  %) were found in the c-axis 19° tilted ScAlN films. Higher  $k_{15}$  value is expected if 30° tilted film is obtained. High k values were observed even in the high temperature of 600 °C.

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Fig. 2 Calculated  $k'_{33}$  and  $k'_{15}$  values of single-crystalline AlN as a function of the c-axis tilt angle. Also plotted are experimentally determined *k* values of the ScAlN films at each c-axis tilt angle determined by pole figure analysis.



Fig. 3 k values as a function of the temperature for the c-axis tilted ScAlN films and c-axis tilted ZnO films and AlN films for a comparison.